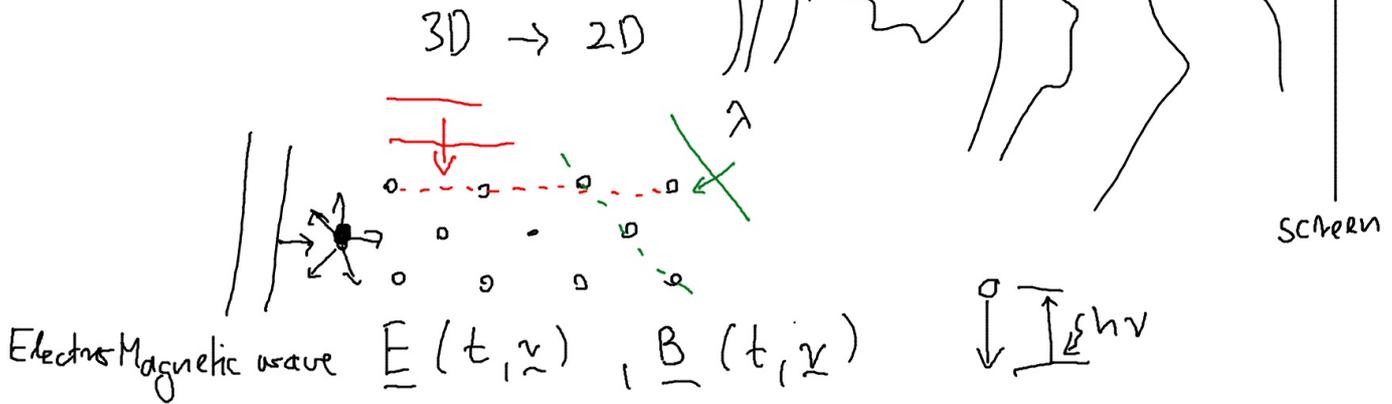


Solid State Physics

Reciprocal lattice



lattice planes
electron beam $\lambda = \frac{h}{p}$

Plane wave \downarrow

$$\underline{E}(t, \underline{r}) = \underline{E}_0 e^{i(\omega t - \underline{k} \cdot \underline{r})}$$

$$E(t, x) = E_0 \sin(\underbrace{\omega t - kx}_{\text{phase}})$$



$$t_0 \exists \quad \omega t_0 - kx_0 = 0$$

$$\boxed{\omega t - kx = 0}$$

\underline{k} direction the w. travels
 $|\underline{k}| = \frac{2\pi}{\lambda}$ wave vector

$$x = \frac{\omega}{k} t$$

$$\underline{R} = \sum_{i=1}^3 n_i \underline{a}_i$$

$$n_i = 0, \pm 1, \pm 2, \dots$$

$$e^{i\varphi} = \cos\varphi + i \sin\varphi$$

$$e^{i(\omega t - \frac{\underline{k} \cdot \underline{r}}{\hbar})}$$

n_1, n_2 equivalent point, i.e.

$$\underline{r}_1 \quad \underline{r}_2 = \underline{r}_1 + \underline{R}_{12} \quad \rightarrow \quad e^{i\frac{\underline{k} \cdot \underline{r}_1}{\hbar}} = e^{i\frac{\underline{k} \cdot (\underline{r}_1 + \underline{R}_{12})}{\hbar}}$$

$$e^{i\frac{\underline{k} \cdot \underline{r}_1}{\hbar}} = e^{i\frac{\underline{k} \cdot \underline{r}_1}{\hbar}} \cdot e^{i\frac{\underline{k} \cdot \underline{R}_{12}}{\hbar}}$$

$$\text{if } e^{i\frac{\underline{k} \cdot \underline{R}}{\hbar}} = 1 \Rightarrow \boxed{\underline{k} \cdot \underline{R} = 2\pi} \quad \infty \quad e^{i2\pi} = 1$$

Δ = distance of lattice planes

Let $\underline{R} = \underline{a}_1 \rightarrow \underline{b}_1 \cdot \underline{a}_1 = 2\pi$ ($\underline{b}_1 \equiv K$ for \underline{a}_1)

$\underbrace{\underline{a}_1, \underline{a}_2, \underline{a}_3}_{\text{direct lattice}} \leftrightarrow \underbrace{\underline{b}_1, \underline{b}_2, \underline{b}_3}_{\text{reciprocal lattice}}$ where $\underline{b}_n \cdot \underline{a}_n = 2\pi$

then

$\underline{b}_1 = 2\pi \frac{\underline{a}_2 \times \underline{a}_3}{\underline{a}_1 (\underline{a}_2 \times \underline{a}_3)}$, but $\underline{a}_1 (\underline{a}_2 \times \underline{a}_3) = \underline{a}_2 (\underline{a}_3 \times \underline{a}_1) = \underline{a}_3 (\underline{a}_1 \times \underline{a}_2)$
 $\underline{b}_2 = 2\pi \frac{\underline{a}_3 \times \underline{a}_1}{\underline{a}_2 (\underline{a}_3 \times \underline{a}_1)}$, notation $\underline{a}_1, \underline{a}_2, \underline{a}_3 \leftarrow$ volume of cell
 $\underline{b}_3 = 2\pi \frac{\underline{a}_1 \times \underline{a}_2}{\underline{a}_3 (\underline{a}_1 \times \underline{a}_2)}$

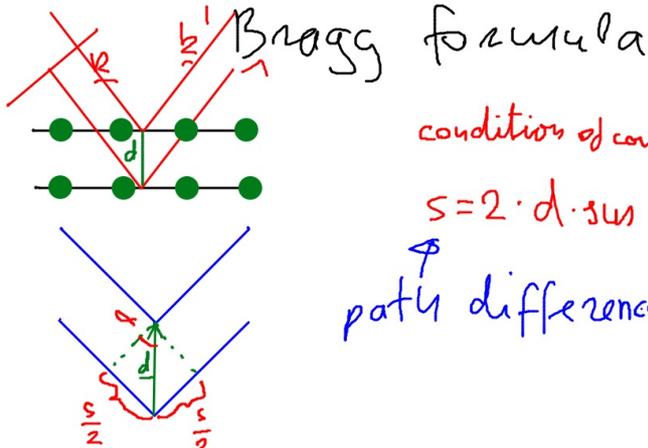
$\underline{R} = n_1 \underline{a}_1 + n_2 \underline{a}_2 + n_3 \underline{a}_3$

$\underline{K} \underline{R} = 2\pi \cdot \underbrace{(n_1 h + n_2 k + n_3 l)}_{\text{integer}}$

$\underline{K} = h \underline{b}_1 + k \underline{b}_2 + l \underline{b}_3$

lattice planes in reciprocal lattice

plane wave
phase is constant



condition of constructive interference:

$s = 2 \cdot d \cdot \sin \alpha = n \cdot \lambda$

path difference

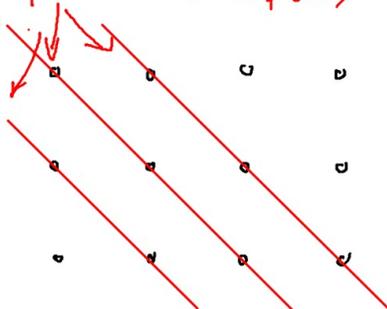
$\underline{K} = \frac{2\pi}{\lambda}$

h, k, l

$\underline{g}_{hkl} = h \underline{b}_1 + k \underline{b}_2 + l \underline{b}_3$

determines a lattice plane

equivalent lattice planes



$h, k, l = 0, \pm 1, \dots$

infinite number of equivalent lattice planes

Miller indices (indexes ☺)

describes all equivalent lattice planes in direct lattice and directions in reciprocal lattice

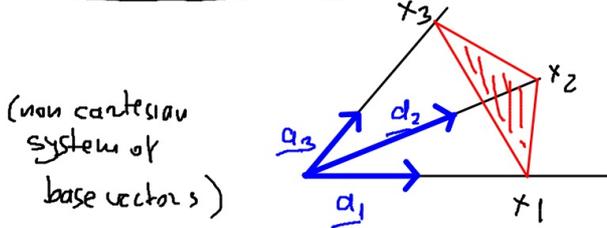
Direction in rec. l. $g_{hkl} = h\underline{b}_1 + k\underline{b}_2 + l\underline{b}_3$ notation: $[hkl]$
equivalent dir.: $\langle hkl \rangle$

Lattice planes.
notation: (hkl) example $(123) \rightarrow h=1, k=2, l=3$

$(1\bar{2}3) \rightarrow h=1, k=-2, l=3$

if any of these is > 9 use commas

method of determination:



$$\begin{aligned} x_1 &= \alpha a_1 \\ x_2 &= \beta a_2 \\ x_3 &= \gamma a_3 \end{aligned}$$

α, β, γ integers, or ∞ if no intersections with the corresponding axis

take the reciprocals: $\left(\frac{1}{\alpha} \mid \frac{1}{\beta} \mid \frac{1}{\gamma} \right)$

using the lowest common denominator A:

then $\frac{1}{\alpha} = \frac{h}{A}, \frac{1}{\beta} = \frac{k}{A}, \frac{1}{\gamma} = \frac{l}{A} \Rightarrow$ use the numerators only:
 $(h, k, l) \Rightarrow (hkl)$

if a plane goes through the origin select another equivalent plane.

(can't divide by 0) $\{hkl\} \leftarrow$ each equivalent numbers for this example:

(001) equivalent with (100) and (010)

so $\{100\}$ means any of these

$$\frac{1}{1} \mid \frac{1}{\infty} \mid \frac{1}{1}$$

(101)

Problem 37. *in a unit cell*
Draw all 9 lattice planes and determine the Miller indexes in a simple cubic Bravais lattice.
Solution:

$\alpha = \infty$	α_3	α_2	α_1	$\{100\}$
$\beta = \infty$				
$\gamma = 1$	β	α	γ	$\{101\}$
$\alpha = 1$				
$\beta = \infty$	α	β	γ	$\{111\}$
$\gamma = 1$				

(Where is the origin of the 3 lattice vectors in the cubes?)

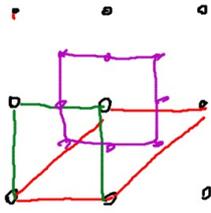
$$\{1\bar{1}1\}$$

special cell in reciprocal lattice:

primitive
primitive

$$h_1, h_2, h_3$$

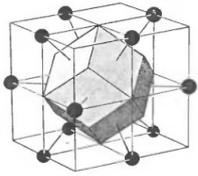
unit cells



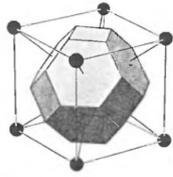
primitive
cell

Wigner-Seitz cell

Or
Brillouin-zone



fcc



bcc

direct lattice

reciprocal lattice

sc \rightarrow

sc

fcc \rightarrow

bcc

bcc \rightarrow

fcc

$g_{hkl} \perp (hkl)$

\uparrow
select the shortest one

$$|g_{hkl}| = d_{hkl} = \sqrt{h^2 a_1^2 + k^2 a_2^2 + l^2 a_3^2}$$

$$\text{distance} = \sqrt{h^2 \left(\frac{2\pi}{a_1}\right)^2 + k^2 \left(\frac{2\pi}{a_2}\right)^2 + l^2 \left(\frac{2\pi}{a_3}\right)^2}$$

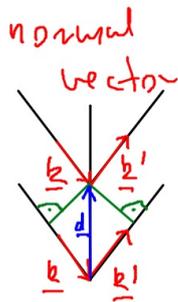
$$|g_{hkl}| = 2\pi \sqrt{\left(\frac{h}{a_1}\right)^2 + \left(\frac{k}{a_2}\right)^2 + \left(\frac{l}{a_3}\right)^2}$$

reciprocal lattice is the Fourier transform of
the direct lattice

Bragg's formula works - why

crystal = point lattice + basis

Laue-formulas



$$\underline{n} := \frac{\underline{k}}{|\underline{k}|} \quad \underline{n}' := \frac{\underline{k}'}{|\underline{k}'|} = \frac{\underline{k}'}{|\underline{k}|}$$

$$s = d(\underline{n}' - \underline{n}) = m \cdot \underline{\Lambda}$$

$$d(\underline{k}' - \underline{k}) = 2\pi \cdot m$$

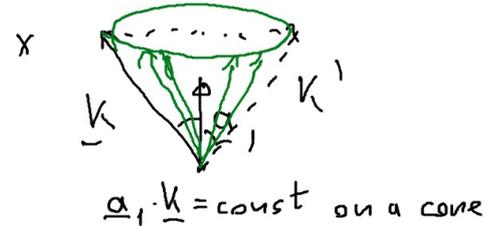
$$\Delta \underline{k} \equiv \underline{K} \quad |\underline{k}| = \frac{2\pi}{\Lambda}$$

for any \underline{k} , even for $\underline{k} = \underline{a}_1, \dots$

$|\underline{k}| = |\underline{k}'|$
elastic process

Laue cond.

$$\begin{matrix} \underline{a}_1 & \underline{k} = 2\pi \underline{h} \\ \underline{a}_2 & \underline{k} = 2\pi \underline{k} \\ \underline{a}_3 & \underline{k} = 2\pi \underline{l} \end{matrix}$$



Experimental methods

ATM, AFM + E.M. waves

what kind of E.M. X-rays ($\lambda \sim nm$)

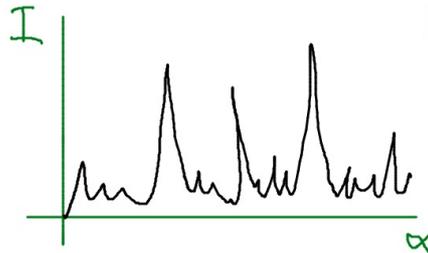
what can we vary?
 Λ , orientation

1. use Δ , λ every $\Lambda < \Lambda_2$
fixed orientation - films
diffraction curves for each Δ

Laue-method

orientation of known

2. Λ and rotate crystal
for Δ



identify unknown crystals

3. powder (microcrystals)
fixed Λ

Debye - Scherrer

lattice parameters